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Improvement of dye-sensitized solar cell performance through electrodepositing a close-packed TiO₂ film

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Abstract A simple electrodepositing method was proposed for fabricating a uniform, tight, and close-packed TiO₂ nanocrystalline film on the ITO substrate. The electrode and dye-sensitized solar cell (DSSC) with electrodeposited TiO₂ layer were characterized by scanning electron microscopy, cyclic voltammetry, and electrochemical impedance spectroscopy. The phthalocyanin dye, zinc tetra-carboxyl phthalocyanin complex, was used as a model dye to evaluate the influence of close-packed TiO₂ blocking layer on the photocurrent-voltage property. On the electrodeposition, the close-packed TiO₂ layer could effectively inhibit the recombination of charges, and therefore improve the performances of the corresponding cells. The effects of film thickness on light transmittance and photocurrent density of the corresponding cell were also demonstrated. The optimum film thickness was found to be approximately 400 nm. At the optimum thickness, the photocurrent density greatly increased comparing with that of the screen printing TiO₂ film. These results imply that our proposition was a potential and feasible method for the fabrication of DSSC practically.

Keywords TiO_2 nanocrystalline film · Electrodeposition · Photocurrent · Electrochemical impedance spectroscopy · Zinc tetra-carboxyl phthalocyanin complex

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Introduction

Recently, dye-sensitized solar cell (DSSC), a device that converts sunlight into electricity on the base of dyesensitized TiO₂ nanocrystalline film, has attracted considerable attention due to its low cost, widely available and biocompatible material, and wide band gap [1, 2]. In a DSSC, an electron transfer cycle was completed by the following process, i.e., the light induced electron rendered into the conducting band of TiO₂ nanocrystallite, and the oxidized dye was reduced in electrolyte by redox couples, usually as iodide/triiodide (I^{-}/I_{3}^{-}) . It was reported that the efficiency of photogenerated electron transition could be improved by using TiO₂ nanocrystallites [1, 3, 4]. A network and transparent mesoporous structure formed by TiO₂ nanocrystallites were ideal for dye loading, thus the sufficient light absorption was benefited. Even though, how to decrease the back reaction of photo-injected electrons with I_3^- was still a challenge.

The opened three-dimensional mesoporous TiO₂ film should favor the electrolyte to penetrate the film to the back contact and induce the dark current at the boundary of conductive substrate. Hence, for enhancement of the DSSC efficiency, it was essential to avoid the recapture of photoinjected electrons by I₃, thereby to prevent charges from recombination. Previous investigations demonstrated that it was effective to introduce a compact and thin layer of TiO₂ underlayer onto the transparent conducting glass substrate, which resulted in the decrease of dark current and enhancement of the DSSC efficiency [5-10]. Three methods were applicable for fabricating such compact TiO_2 film, namely spray pyrolysis [5, 7], electrophoretical deposition [6], and dip coating [9]. Spray pyrolysis deposits a compact TiO₂ underlayer followed by coating a mesoporous TiO_2 layer with screen printing [5, 7]. Grätzel and

co-workers [7] invented it for the suppression of dark current so as to increase $V_{\rm OC}$. Peng [8] obtained a compact TiO₂ layer with the best photovoltaic properties at an optimum film thickness of 150 nm corresponding to ten cycles spraving by using an aerosol of titanium (IV) bis (acetoacetonato) di(isopropanoxylate) (TAA) in ethanol as the spraving substrate. However, the disadvantages of sprav pyrolysis were obvious. For example, high temperature was essential to pyrolyze the organic substance into gas state; in addition, multi-step and time-costing operation with more than ten spraying cycles was also disadvantageous. Yum et al. [6] developed the electrophoretical deposition. They reported that the current density was increased obviously, with its low cost, relatively fast and reproducible, whereas the improvement in interconnection between nanocrystalline TiO₂ by compression was needed. Hart et al. [9] declared that a compact film of TiO₂ sol-gel blocking layer was constructed by dip-coating method. It increased the short-circuit current, but the operation was complicated and the thickness of film was uncontrollable. Anyway, these studies illustrated that the dark current charge recombination occurred mainly near the conductive substrate and not across the entire TiO_2 film; thus, the formation of TiO_2 underlayer on a conductive substrate could effectively block the dark current.

In the present work, which targeted to overcome all those demerits of the above methods, an electrodepositing method was developed for preventing the loss of photoinjected electrons from the back reaction in DSSC solar cell. It was simple and economic, compared with the above methods.

Experimental

Materials

TiCl₃ was obtained from Sigma Aldrich Co. Ltd. (USA). Tetrabutylorthotitanate ($C_{16}H_{36}O_4Ti$) was purchased from Lingfeng Chemical Co. Ltd. (Shanghai, China). Tetrabutylammonium perchlorate ($C_{16}H_{36}NClO_4$, TBAP) was obtained from Xianju Medicine and Chemical Co. Ltd. (Zhejiang, China). The colloidal TiO₂ solution for screen printing was prepared by dropwise adding (one drop per 5 s) a mixture of 12.5 ml tetrabutylorthotitanate and 4 ml isopropanol into 150 ml H₂O containing 2 ml dense HNO₃. The solution was then kept at 75 °C for 5 h after adding PEG800 into the solution so as to avoid the aggregation of TiO₂ particles. TiO₂ colloid was thus obtained and stored at room tempeture for the following use.

Zinc tetra-carboxyl phthalocyanin complex was synthesized according to the reference [11] with slight modification as shown in Scheme 1. Briefly, mixing 6.8 g trimellitic anhydride, 2.2 g zinc acetate, and 12.0 g urea at first and then adding 0.02 g ammonium molybdate as catalyst, the mixture was heated to 200 °C for the high yield of intermediate. Thereafter, it was refluxed in potassium hydroxide for 20 h at 100 °C and washed with 2.0 M hydrochloric acid for several times. The yield of ZnTCPc was ca. 1.02 g.

The resistance of ITO conductive glass was $15\Omega/sq$. All other chemicals were of analytical grade and used without further purification.

Systhesis of TiO₂ close-packed layer on ITO electrodes

Prior to each electrodeposition, the ITO glass was cleaned by soaking in KOCH(CH_3)₂ for 0.5 h to erase the grease. Thereafter, it was ultrasonicated thoroughly with distilled water and ethanol, respectively, and then dried with nitrogen flow.

For preparation of the close-packed TiO₂ layer on ITO electrode surface, the electrolyte was prepared by diluting 8 ml 20-30 wt.% TiCl₃ solution with deoxygenated deionized water to 40 ml and adding 0.1 g of Zn powder in order to prevent Ti³⁺ from the oxidation to Ti⁴⁺, and then kept in dark for 12 h. After adjusting pH to 2.2-2.5 by the saturated NaOH aqueous solution, the electrodeposition was performed, using a conventional three-electrode electrochemical cell equipped with a Pt-wire auxiliary electrode, a saturated calomel electrode (SCE) reference electrode, and an ITO conductive glass $(1 \times 2 \text{ cm}^2)$ as the working electrode at 0.1 V during different time periods. After washing thoroughly with deionized water and dried under infrared lamp, the resulting TiO₂ electrodeposited electrodes, denoted as eTiO2/ITO, was used for further coating a layer of TiO₂ film (about 5µm thickness) by screen printing method. After being annealed at 450 °C for 30 min under ambient conditions, the electrodes with twolayer TiO₂ film (tTiO₂/ITO) were thus obtained. For comparison, the mono-screen-printing electrode (sTiO₂/ ITO) without TiO₂ electrodeposited layer was prepared only by screen printing followed by annealing at 450°C for 30 min under ambient conditions. All the results are reproducible.

The morphologies of the nanocrystalline TiO_2 film electrodes were characterized by scanning electron microscopy (SEM, LEO 1530 VP, Germany). Gold was used as the sputtering material.

Zinc tetra-carboxyl phthalocyanine-sensitized TiO₂ film

Piror to sensitization, TiO_2 particles was pretreated by immersing TiO_2 film-coated electrodes into pH 3.0 HCl solution for 12 h. The electrode was washed thoroughly

Scheme 1 The synthesis of zinc tetra-carboxyl phthaocyanine (ZnPcTc)



with water and dried at atmosphere. After heating at 120° C for 2 h, the electrode was soaked in 0.5 mM DMF solution of ZnTCPc for 24 h. It was then rinsed thoroughly with DMF and dried at 120° C for 2 h to form a dye-sensitive TiO₂ film electrode.

Measurements

The UV–vis transmittance spectra of TiO₂ film electrode were performed with a UV-2201 spectrophotometer (Hitachi Shimadzu, Japan). Electrochemical measurements were carried out with a CHI 750C electrochemical station. Cyclic voltammetric experiments and electrochemical impedance spectroscopy were recorded by using a three-electrode system, a nanocrystalline TiO₂ film working electrode (geometric area of 25 mm²), a platinum auxiliary electrode, and a saturated sodium calomel electrode (SCE) reference electrode. The electrolyte was 2.5 mM 1:1 K₄Fe(CN)₆/K₃Fe (CN)₆ in 0.1 M KCl solution. All experimental solutions were deoxygenated by bubbling highly pure nitrogen for 30 min, and maintained under nitrogen atmosphere during measurements.

Photocurrent–voltage characteristics of dye-sensitized TiO_2 film electrodes were measured with sandwich-type cells. Dye-sensitized TiO_2 film was used as a working electrode, while Pt sputtered on ITO was used as a counter electrode. The electrolyte was 0.5 M KI+0.05 M I₂ in propylene carbonate. Photocurrent–photovoltage characteristics of solar cells were measured by a Keithley model 2400 digital source meter using a NEWPORT Oriel 91192 solar simulator (NEWPORT, USA) equipped with AM 1.5 filter and the intensity was adjusted to 100 mW/cm².

Results and discussion

The preparation of TiO₂ film

It has been reported that the formation of TiO_2 underlayer on a conductive substrate could effectively block dark current, and thus being widely used in DSSC design [5– 10]. In the present work, a close-packed eTiO₂/ITO prepared by electrodeposition was applied. For comparison, sTiO₂/ITO was also constructed. Figure 1 showed the SEM images of top-view TiO_2 layer on ITO substrate fabricated by electrodeposition at 0.1 V for 20 min (a) and by screen printing method (b). The average particle size estimated from the SEM image was about 20 nm (Fig. 1a), whereas it was approximately 30 nm by screen printing methods (Fig. 1b). Meanwhile, as shown in Fig. 1a, the electrodeposited TiO₂ layer had a densely packed and homogeneous structure, whilst porous morphology was observed for sTiO₂/ITO. It indicated that a compact layer of TiO₂ film on ITO substrate was achieved by electrodeposition.

Furthermore, the film thicknesses of close-packed TiO₂ on ITO electrode surface were determined by SEM. The close-packed TiO₂ layers were deposited in 20-30 wt.% TiCl₃ solution at 0.1 V for (a) 15, (b) 25, and (c) 35 min, respectively. As shown in Fig. 2, after electrodepositing for 15 min, a flat and compact layer of TiO₂ film with thickness of 200 nm was grown on the top of ITO surface. The film thickness increased with the increase of electrodeposition time (Fig. 2b, c). For example, as the depositing time was prolonged from 25 to 35 min, it increased from 320 to 460 nm. However, it should be noted that the heterogeneously distributing film due to the layer stack of TiO₂ nanoparticles was observed when the depositing time was 35 min (not shown), and further studies showed that the performance of solar cells with such heterogeneous TiO_2 film was poor. Therefore, it was concluded that the film thickness should be controlled at less than 400 nm, and that the depositing time should be around 25 min. Hereby, a flat and compact TiO₂ film with a film thickness of approximately 400 nm was achieved. This film, applied for the following studies, was expected to prevent the electrolyte from diffusing through the film to reach the back contact of conductive substrate and in turn to reduce the recombination of charges.

The formation of close-packed TiO₂ film on the surface of ITO by electrodepositing was also evidenced by electrochemical results. The redox couple $Fe(CN)_6^{3-/4-}$ was widely used as an electrochemical probe in the electrochemical study, especially for the characterization of film-modified electrodes. Using K₃[Fe(CN)₆]/K₄[Fe (CN)₆] as probe, as shown in Fig. 3a, it displayed a pair of well-defined redox peaks of $Fe(CN)_6^{3-/4-}$. After a compact TiO₂ layer was coated, poor redox peaks with a peak-to-peak separation potential of 565 mV were observed Fig. 1 The SEM images of a top-view TiO_2 layer on ITO fabricated by a electrodeposition at 0.1 V for 20 min, b by screen printing method



for $eTiO_2/ITO$ (Fig. 3b). Moreover, the peak currents of $sTiO_2/ITO$ showed slight decrease, comparing with those of bare ITO electrode (Fig. 3c). These results indicated that, as expected, the close-packed TiO_2 nanocrystalline film on ITO effectively blocked the electron transfer from ferricy-anide to the electrode. Additionally, it prevented ferricyanide from penetrating the film, thereby blocking the exchange of electron with electrode. As a result, it decreased the electrochemical response of electroactive probe.

The results of electrochemical impedance spectra further supported the formation of densely packed TiO₂ layer by electrodeposition. As we know, impedance spectroscopy was an effective approach to probe the interfacial properties of surface-modified electrode [12-17]. The complex impedance was donated as the real, $Z_{\rm re}$, and imaginary, $Z_{\rm im}$, components mainly originating from the resistance and capacitance of cell, respectively. Figure 4a showed the AC impedance spectra for (a) bare ITO substrate, (b) electrodeposition, (c) screen printing, and (d) electrodeposition following screen printing TiO₂ film electrode in 0.1 M KCl solution containing 2.5 mM $Fe(CN)_6^{3-/4-}$ at an open circumpotential. An equivalent circuit $R_s(R_{ct}CPE)W_o$ (Fig. 4b) was used to model the impedance data, thus enabling the extraction of electrical parameters, such as resistance, from the impedance spectra [18]. In this circuit, $R_{\rm s}$ represented the ohmic resistance of electrolyte, while the parallel combination of charge-transfer resistance (R_{ct}) appeared when a redox probe existed in the electrolyte solution and the constant phase element (CPE) led to a depressed semicircle in the corresponding Nyquist impedance plot, and W_{0} was the Warburg impedance resulted from the diffusion of ions from the bulk electrolyte to electrode. CPE was defined by CPE-T and CPE-P. If CPE-P was 1, then CPE was considered as a capacitor C_{dl} . The components R_s and W_o in the electronic circuit represented bulk properties of electrolyte solution and diffusion features of redox probe in solution. The other two components in circuit, CPE and R_{ct} , represented the interfacial properties of electrode and they were affected by coating the nanostructured materials onto the electrode surface. Therefore, the analysis of CPE and R_{ct} could give important information of the varying extent of surface properties.

It was expected to introduce a barrier for the electron transfer at the electrode interface by coating TiO₂ film on the electrode surface. As a result, an enhanced electron transfer resistance (R_{ct}) was observed from 171.4 Ω for the bare ITO electrode to 476.0 Ω for sTiO₂/ITO and 591.4 Ω for eTiO₂/ITO, respectively. The reason for the increase in $R_{\rm ct}$ for eTiO₂/ITO relative to sTiO₂/ITO was considered that the denser packed TiO₂ nanoparticles on the electrode surface stopped the ferricyanide from exchanging electron with electrode, thereby effectively blocking the reactions of redox active species at electrode. R_{ct} of eTiO₂/ITO further increased from 591.4 to 812.1 Ω by coating a mesoporous layer on the electrodeposited TiO₂ film. It possibly resulted from the accumulative effect of impedance at both electrodepositing and screen printing TiO₂ layer. Hence, the introduction of densely packed TiO₂ layer between ITO substrate and mesoporous TiO₂ layer enhanced the power conversion efficiency of DSSC by preventing the electrolyte from bypassing the film and reducing charge recombination at the boundary of conductive substrate.

The effect of film thickness on light transmittance and photocurrent

The close-packed TiO₂ film was transparent, which transmitted more than 90% of injected lights. Figure 5 showed spectra of UV–vis transmittance of TiO₂ film electrode prepared by electrodeposition at 0.1 V for 20 min referencing a bare ITO. The transmittance of deposited TiO₂ film increased with the red shift of wavelength and leveled off in the range of wavelength longer than 581 nm. The largest transmittance considerably favored phthalocyanine dyes of TCZnPc to gain the highest absorbance of Q



Fig. 2 Cross-section SEM images of the close-packed TiO_2 layers prepared by electrodepositing at 0.1 V for a 15, b 25, and c 35 min

band between 600 and 800 nm, thus the high energy conversion efficiency was expectable.

The light transmittance of $eTiO_2/ITO$ was dependent to the thickness of electrodepositing TiO_2 film. By varying the depositing time, a series of samples with increased thickness was obtained. It was observed that the transmittance of various films at 358 nm decreased from 96% to 20% as the depositing time prolonged (Fig. 6). Moreover, the thickness can be readily controlled by changing deposition parameters such as deposition time.



Fig. 3 Cyclic voltammograms of **a** bare ITO substrate, **b** electrodeposition, and **c** screen printing TiO_2 film electrode in 0.1 M KCl solution containing 2.5 mM K₄Fe(CN)₆/K₃Fe(CN)₆

Further investigation was carried out by screen printing a mesoporous TiO₂ layer on the above $eTiO_2/ITO$. Accordingly, the $tTiO_2/ITO$ with different thickness of close-packed TiO₂ underlayer was prepared. After being annealed and immersed in HCl solution, the $tTiO_2/ITO$ was soaked in 0.5 mM ZnTCPc for 24 h to obtain dye-sensitized TiO₂ film electrodes. Equipped with a dye-sensitized TiO₂ film electrode as working electrode, ITO sputtered by Pt as counter electrode, sandwich cells were employed to evaluate the photocurrent–voltage performances of dye-sensitized TiO₂ film electrodes. Figure 5 displayed the plots of related photocurrent density versus film thickness. The



Fig. 4 a Nyquist plots of (*a*) bare ITO substrate, (*b*) electrodeposition, (*c*) screen printing, and (*d*) electrodeposition following screen printing TiO₂ film electrode in 0.1 M KCl solution containing 2.5 mM Fe(CN)₆^{3-/4-} at an open circum potential. **b** An equivalent circuit $R_{\rm s}(R_{\rm ct}{\rm CPE})W_{\rm o}$ used to model the impedance data



Fig. 5 The transmittance spectra of the close-packed TiO_2 film on ITO substrate by electrodeposition at 0.1 V for 20 min. Bare ITO used as reference

related photocurrent density, donated as $R_{\rm JSC}$, was calculated by the following formula:

$R_{\rm JSC} = J_{\rm SC, tTiO_2/ITO} / J_{\rm SC, sTiO_2/ITO}$

Where $J_{SC,tTiO_2/ITO}$ is the photocurrent density of $tTiO_2/ITO$ measured in the corresponding DSSC and $J_{SC,sTiO_2/ITO}$ the photocurrent density of $sTiO_2/ITO$ without the underlayered close-packed TiO₂ film. The related photocurrent density increased as the thickness of the deposited TiO₂



Fig. 6 a The effect of electrodeposition time on transmittance of the close-packed TiO_2 film at 358 nm using bare ITO as a reference. **b** Plots of related photocurrent density vs. electrodeposition times

compact layer increased and reached a maximum at the electrodepositing time, 25 min. It was reported that the increase in photocurrent density was mainly due to the effective inhibition of recombination of charges, which led to more electrons injecting from dve to the ITO anode, via the mesoporous TiO2 and close-packed TiO2 blocking film [19]. The thicker the close-packed TiO₂ films, the more effective they block the holes produced from getting through to the anode and, therefore, resulting in an increase in photocurrent density. By further increasing the film thickness, however, the related photocurrent density decreased. The much thicker close-packed TiO₂ (e.g., electrodepositing time, 35 min, and the thickness, 460 nm) could hinder the desired number of electrons reaching the ITO anode due to its surface defaults and poor light transmittance as discussed above [19]. Therefore, an optimum thickness, 400 nm, of the blocking TiO₂ layer, corresponding to electrodepositing time, 25 min, was appreciated to balance the blocking of dark current recombination and transportation of electrons.

At the optimum thickness of 400 nm, the photovoltage $(V_{\rm OC})$, photocurrent density $(J_{\rm SC})$, and the fill factor (FF) of tTiO₂/ITO-based cell were 0.416 V, 1.19 mA/cm², and 0.46, respectively (Fig. 7). According to these results, it was concluded that the performance was better than that of sTiO₂/ITO-based cell (0.363 V, 0.585 mA/cm², and 0.49). In other words, the close-packed TiO₂ layer was effective to reduce the dark current, thereby to improve the performance of DSSC.

Conclusions

A uniform, tight, and close-packed TiO_2 nanocrystalline film on the ITO substrate was fabricated by an electrodepositing method. This compact layer could effectively inhibit the recombination of charges and led to more electrons injecting from dye through the mesoporous TiO_2



Fig. 7 Photoelectric characterization of a ZnPcTc-sensitized TiO_2 film solar cell with **a** sTiO₂/ITO, **b** tTiO₂/ITO (0.1 V, 25 min)

and close-packed TiO₂ blocking film to the ITO anode, and therefore improved the performances of the corresponding cells. The effects of film thickness on light transmittance and photocurrent density of the corresponding cell were also demonstrated. Considering the largest related photocurrent density and good light transmittance, the optimum film thickness was found to be approximately 400 nm. Using this optimum thickness film, the photocurrent density greatly increased comparing with that of the screen printing TiO₂ film. The improvements of the photoelectrical characteristics made the proposed method potential to be used for practical fabrication of DSSC.

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